

Short communication

# Nanostructured manganese oxide electrodes for lithium-ion storage in aqueous lithium sulfate electrolyte

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## Abstract

Nanostructured manganese oxide electrodes are fabricated directly by electrochemical deposition. Surface morphology of the electrode deposited at high-current density shows nanowires with diameter 12–16 nm distributed randomly. Nanowires tend to aggregate to clumps when the deposition current density is low. Both annealing temperature and deposition current density affect the electrochemical performance of the deposited manganese oxide electrode in an aqueous lithium sulfate electrolyte. An optimal annealing temperature is found to be 300 °C in terms of the electrode's specific capacity during high-rate charging/discharging. An electrode with thinner nanowires deposited at high-current density has a high-specific capacity because thinner nanowires shorten the diffusion of lithium ions and in favor of high-rate charging/discharging.

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**Keywords:** Manganese oxide; Lithium-ion storage; Electrochemical capacitors; Nanostructured materials; Aqueous lithium-ion batteries

## 1. Introduction

There are many different forms of manganese dioxide such as  $\alpha$ -MnO<sub>2</sub>,  $\beta$ -MnO<sub>2</sub>, and  $\gamma$ -MnO<sub>2</sub> for lithium battery cathode material. Generally, crystal structure of the manganese oxides influences intercalation and deintercalation of lithium ions during electrochemical reaction.  $\alpha$ -MnO<sub>2</sub> has low-electrochemical stability because it contains (2 × 2) tunnels within its structure [1–4]. To enhance the structural instability of  $\alpha$ -MnO<sub>2</sub>, cations such as Ba<sup>2+</sup> or K<sup>+</sup> were added during synthesis because they can take up crystallographic sites within the (2 × 2) tunnels [4]. Although structural stability of  $\alpha$ -MnO<sub>2</sub> has been improved by cation addition, a drawback is that electrochemical performance would be significantly limited by a cation-hindered and hence lowered diffusion rate of lithium ions [4,5].

$\beta$ -MnO<sub>2</sub> structure is more stable in the manganese dioxide family because its structure consists of narrow (1 × 1) tunnels [4]. However, this stable structure obstructs the intercalation and deintercalation of lithium ions, therefore results a low capacity [6]. Specific capacity of  $\beta$ -MnO<sub>2</sub> can be much improved by

decreasing the degree of crystallinity, but the charge/discharge reaction cannot be fully reversed [7,8].

$\gamma$ -MnO<sub>2</sub> structure consists of intergrown domains of  $\beta$ -MnO<sub>2</sub> and ramsdellite-MnO<sub>2</sub> in which the ramsdellite-MnO<sub>2</sub> embeds (2 × 1) tunnels [4,9]. Reports have stated the reversible intercalation and deintercalation of lithium ions occur mainly in the ramsdellite-MnO<sub>2</sub> domain of  $\gamma$ -MnO<sub>2</sub> [10,11]. Generally, chemical or electrochemical synthesized  $\gamma$ -MnO<sub>2</sub> material contains water on the grain surface or grain boundary. Water can be removed by heating the substance to 375 °C [12].

Nanostructured materials play an important role in electrochemical performance because high-specific surface area and fast redox reactions enhance electrochemical behavior of the applied battery. Generally, lithium-ion diffusion within the crystal structure of active material dominates the high-rate charging and discharging of the electrode. Diffusion resistance of lithium ions within the material can be decreased by shortening the diffusion path; therefore, nanostructure is advantageous in improving the high-rate performances of materials.

However, an electrode composed of nanoparticles is more difficult to fabricate by the traditional slurry coating method, because nanoparticles have poor dispersibility in slurry (composed of solvent, nanoparticles, polymer binder, conducting agent, etc.). West et al. [13] have proposed a synthesis method for preparing the manganese oxide array by deposition of the

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manganese oxide sol–gel within the porous template. Sugantha et al. [14] deposited the manganese oxide in a porous template to form array electrode and demonstrated a much improved high-rate performance. Using a template method to synthesize nanostructured materials, the template will need to be removed after synthesis. Thus, it is more advantageous to have an electrode of nanosized manganese oxide fabricated by electrochemical deposition which deposits the active material directly onto the substrate at room temperature without any template and/or catalyst. Electrochemical deposition technique has one advantage over all the others: weight and thickness of the metal oxide film may be easily controlled by controlling the current, bath composition, and bath temperature.

Most of the researches on lithium-ion batteries are focused on the nonaqueous electrolyte. Recently, Li et al. [15–17] have proposed a type of rechargeable lithium batteries with aqueous electrolyte. Aqueous lithium-ion battery is one of the promising candidates for energy storage in terms of safety and cost. More recently, Wang et al. [18–20] have proposed a hybrid aqueous energy storage cell using  $\text{LiMn}_2\text{O}_4$ ,  $\text{LiCoO}_2$ , and  $\text{LiCo}_{1/3}\text{Ni}_{1/3}\text{Mn}_{1/3}\text{O}_2$  as the positive electrode and activated carbon as the negative electrode. Therefore, in this work, the electrochemically anodic deposition method is used to fabricate manganese oxide directly onto a stainless steel substrate. In addition, electrochemical performance of the synthesized manganese oxide is investigated in an aqueous electrolyte.

## 2. Experimental

Nanostructured manganese oxide electrodes were electrochemically deposited onto stainless steel (SS) foils ( $2\text{ cm} \times 2\text{ cm}$ ) by applying two current densities of  $0.125\text{ mA cm}^{-2}$  and  $0.025\text{ mA cm}^{-2}$ , respectively. The plating solution consisted of  $0.1\text{ M}$  manganous acetate and  $0.1\text{ M}$  sodium sulfate at room temperature [21–23]. A saturated calomel electrode (SCE) was used as the reference electrode and a platinum foil with dimension  $2\text{ cm} \times 2\text{ cm}$  was used as the counter electrode. Prior to deposition, SS foils were polished with emery paper, and washed in acetone and deionized water, respectively. The plating solution was stirred by a Teflon stir on a magnetic hot plate during the entire deposition. After deposition, the deposited foils were rinsed several times in deionized water and dried at various temperatures for 1 h in air. The amount of deposited manganese oxide was measured by a microbalance (Ohaus G160, USA) with an accuracy of  $0.01\text{ mg}$ .

Surface morphology of the electrochemically deposited electrode was examined with a scanning electron microscope (FE-SEM, Jeol JEOL-6330, USA) with an accelerating voltage of  $15\text{ keV}$ . Crystal structures of the deposited manganese oxide were identified by a glance angle X-ray diffractometer (GAXRD, Rigaku D/MAX2500, Japan) with a  $\text{Cu K}\alpha$  target (wavelength =  $1.54056\text{ \AA}$ ).

A beaker-type electrochemical cell comprised of a working electrode (manganese oxide electrode), a counter electrode (platinum foil), and a reference electrode (SCE) was used to determine the electrochemical performance of deposited man-

ganese oxide electrodes. The electrolyte was  $1\text{ M}$  lithium sulfate ( $\text{Li}_2\text{SO}_4$ ) aqueous solution. Charge/discharge and cycle-life stability were performed by a charge/discharge unit (Hokuto Denko HJ-201B, Japan) in the potential range of  $0\text{--}1.1\text{ V}$  versus SCE at constant current. All data acquisition functions in Hokuto Denko were carried out through an interface card (Labjack U12, USA) with LabVIEW software. Electrodes were charged at constant current to a cutoff potential of  $1.1\text{ V}$  versus SCE. Discharge was performed at the same rate to a cutoff potential of  $0\text{ V}$  versus SCE. Cyclic voltammetry (CV) measurements were taken by means of a potentiostat/galvanostat (CH Instruments CHI 608, USA). The potential was cycled in the range of  $0\text{--}1.2\text{ V}$  versus SCE at a scan rate of  $25\text{ mV s}^{-1}$ .

## 3. Results and discussion

Previous results indicated that the deposited manganese oxides of annealing temperatures lower than  $300\text{ }^\circ\text{C}$  resemble closely to  $\gamma\text{-Mn(O,OH)}_2$  ( $\gamma$ -manganese oxide hydroxide), but it decomposes to  $\text{Mn}_2\text{O}_3$  at temperatures beyond  $500\text{ }^\circ\text{C}$  [24]. In order to avoid structural destruction by water contamination, the annealing temperature has been set at  $300\text{ }^\circ\text{C}$ . Fig. 1 shows the surface morphology of manganese oxide electrode deposited electrochemically at different current densities after annealing at  $300\text{ }^\circ\text{C}$  for 1 h. Morphology of the deposited manganese oxide at a high-current density of  $0.125\text{ mA cm}^{-2}$

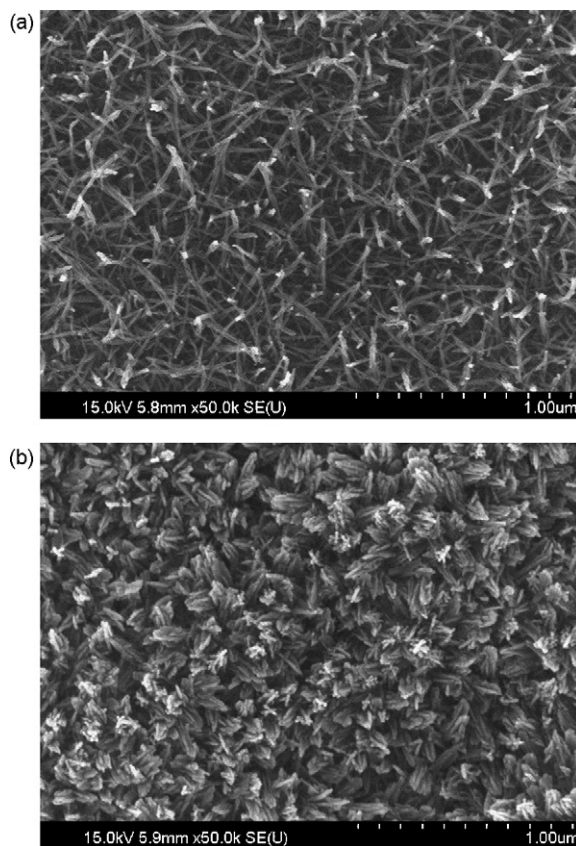


Fig. 1. Surface morphology of the manganese oxide electrode deposited electrochemically at current density of  $0.125\text{ mA cm}^{-2}$  (a) and  $0.025\text{ mA cm}^{-2}$  (b) after annealing at  $300\text{ }^\circ\text{C}$  for 1 h.

(Fig. 1a) shows nanowires with diameter of 12–16 nm and distributes randomly on the SS substrate. Nanowires tend to aggregate into clumps of diameter about 100 nm (Fig. 1b) when current density is decreased to  $0.025 \text{ mA cm}^{-2}$ . Current density of the deposition is important in controlling the surface morphology of the deposited manganese oxide film, which in turns affect its electrochemical performance. It is generally believed that nanostructured materials have higher specific surface area and fast redox reactions; in addition, ion diffusion within the structure dominates high-rate charging and discharging performances of the electrode. Diffusion resistance of ions (lithium ions) within the materials can be mitigated by shortening the diffusion path; therefore, electrode materials with nanosized wires may be advantageous in high-rate performances.

Fig. 2 shows the cyclic voltammograms of the deposited manganese oxide electrodes after annealing at different temperatures: electrolyte is 1 M  $\text{Li}_2\text{SO}_4$  aqueous solution and CV scan rate is  $25 \text{ mV s}^{-1}$ . The potential was cycled between 0 V and 1.2 V versus SCE corresponding to a range of 3.3–4.5 V versus  $\text{Li}/\text{Li}^+$  in nonaqueous electrolyte. When the heating temperature exceeds  $200^\circ\text{C}$ , two distinct redox current peaks may be observed: an anodic peak at ca. 0.9 V and a cathodic peak at ca. 0.75 V versus SCE. Based on previous reports [25,26], appearances of redox peaks indicate a reversible intercalation/deintercalation of lithium ions in solid phase with charge transfer on the electrode/electrolyte interface. It is believed that charge (lithium-ion) storage mechanisms of the manganese oxide in aqueous solution include storage of ions within the electric double layer, adsorption of ions on the surface layer of manganese oxide, and insertion of ions into the host materials through electrochemical faradaic reactions [25–28]. In addition to redox peaks, the CV shape of the electrode is roughly rectangular mirror image with respect to zero-current line indicating that the capacitive behavior is also included. Consequently, capacity of the synthesized manganese oxide comes from stor-

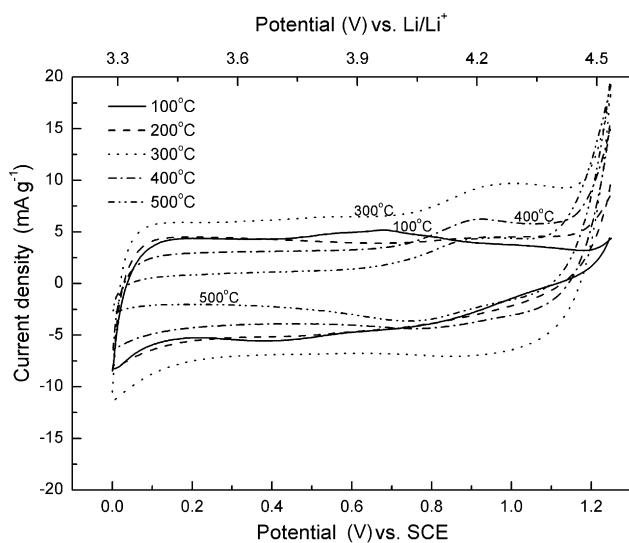


Fig. 2. Cyclic voltammograms of deposited manganese oxide electrodes after annealing at different temperatures for 1 h. Electrodes were deposited at  $0.125 \text{ mA cm}^{-2}$ .

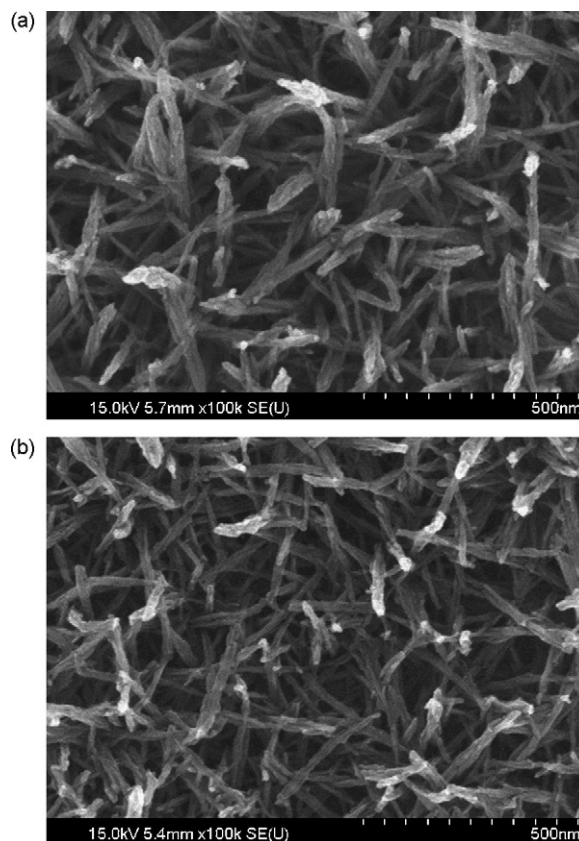


Fig. 3. Surface morphology of the deposited manganese oxide electrode at fully intercalated state (a) and fully deintercalated state (b). Electrode was deposited at  $0.125 \text{ mA cm}^{-2}$  and annealed at  $300^\circ\text{C}$  for 1 h.

age of lithium ions within the electric double layer, adsorption of lithium ions on the surface layer of manganese oxide, and insertion of lithium ions into the host materials through electrochemical faradaic reactions. Electrode after annealing at  $300^\circ\text{C}$  has the highest current response in cyclic voltammogram, indicating the largest storage capacity for lithium ions at this annealing temperature.

Fig. 3 shows surface morphology of the deposited manganese oxide electrode at full intercalated and deintercalated states. Electrode was deposited at  $0.125 \text{ mA cm}^{-2}$  and annealed at  $300^\circ\text{C}$  for 1 h. After intercalation and deintercalation of lithium ions, morphology of nanowires remains almost unchanged except a small volumetric expansion at the deintercalated state as compared with the intercalated state; which is quite different from manganese oxide used in anode in a nonaqueous electrolyte. Previous results showed that surface morphology of manganese oxide electrode treated at  $300^\circ\text{C}$  has changed drastically after charging from 3.0 V to 0.01 V versus  $\text{Li}/\text{Li}^+$  [24,29]. Lithium-storage process in manganese oxide anode is unlike the process in cathode because morphology after lithium intercalation does not change significantly, it only expands a little volumetrically.

Fig. 4 shows the charge (deintercalation) and discharge (intercalation) curves of the deposited manganese oxide electrode at different charging/discharging current densities. Electrode was electrochemically deposited at  $0.125 \text{ mA cm}^{-2}$  and annealed

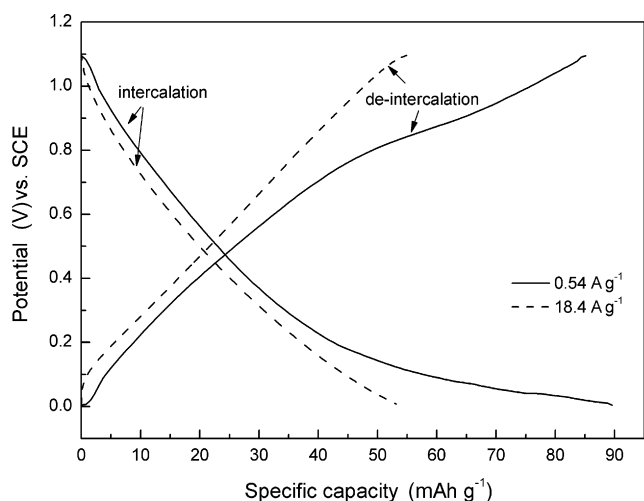


Fig. 4. Charge and discharge curves of the deposited manganese oxide electrode at different charging/discharging current densities. Electrode was deposited at  $0.125 \text{ mA cm}^{-2}$  and annealed at  $300^\circ\text{C}$  for 1 h.

at  $300^\circ\text{C}$  for 1 h. At charging/discharging current density of  $0.54 \text{ A g}^{-1}$ , charge capacity of  $90 \text{ mAh g}^{-1}$  is almost the same as the discharge capacity. With a higher charging/discharging current density of  $18.4 \text{ A g}^{-1}$ , the electrode capacity reaches  $54 \text{ mAh g}^{-1}$ . Specific capacity decreases only slightly by increasing the current, meaning that the synthesized manganese oxide electrode is insensitive and tolerates high-rate charging and discharging. Possibly, this insensitivity results from its nanowire structure: large electroactive area for fast redox reaction, and shortened diffusion paths for high-rate charging/discharging. In general, for a cathode material in lithium-ion batteries, the potential plateau appears at low-current density discharge due to the intercalation of lithium ions. In Fig. 4, the potential plateau is not apparent during discharging except for near the discharging end. This is the reason that the discharging current in the manganese oxide electrode is very high as compared with the discharging current in traditional cathode materials. From Fig. 4, the potential plateau can be found clearly at a lower current density of  $0.54 \text{ A g}^{-1}$  indicating the intercalation of lithium ions into the manganese oxide electrode.

Fig. 5 shows the combined effect of annealing temperature and discharging current density on specific capacity of the electrochemically deposited manganese oxide electrode. The relationships show that annealing temperature influences remarkably the specific capacity of electrode at all charging/discharging current ranges. Specific capacity value increases with annealing temperature, but it reaches a maximum at  $300^\circ\text{C}$ , and decreases with further increase in annealing temperature. The deposited film annealed at  $300^\circ\text{C}$  shows a superior performance in high-rate charge/discharge capability: the film has a specific capacity of about  $70 \text{ mAh g}^{-1}$  at discharging current density  $1.15 \text{ A g}^{-1}$ , and  $54 \text{ mAh g}^{-1}$  at  $18.4 \text{ A g}^{-1}$ .

Previous reports suggested that annealing temperature affects crystal structure of the electrochemically deposited manganese oxide [28,30–32]. According to Preisler [31], only a small percentage of water in the electrodeposited manganese dioxide is volatile at  $120^\circ\text{C}$ , and a predominant portion of water is des-

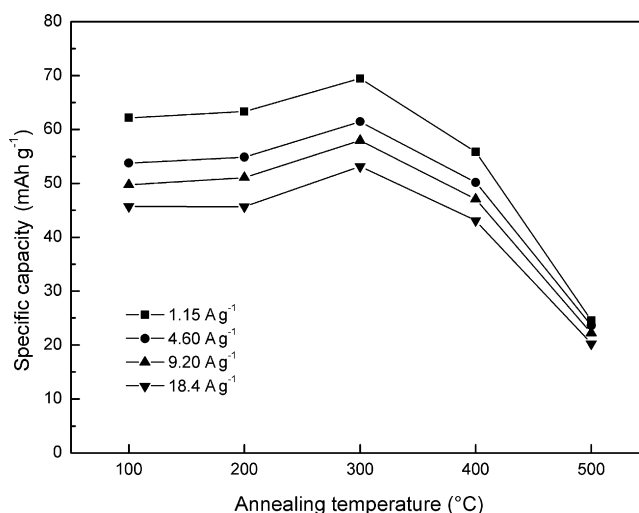


Fig. 5. Effect of annealing temperature and discharging current density on the specific capacity of the electrochemically deposited manganese oxide electrode. Electrodes were deposited at  $0.125 \text{ mA cm}^{-2}$ .

orbed in a relatively smooth manner up to  $350^\circ\text{C}$ . In addition, the electrodeposited manganese oxide decomposes rapidly to form  $\text{Mn}_2\text{O}_3$  at temperatures exceeds  $500^\circ\text{C}$  [30]. Water content within is known to affect the electrochemical reactivity and the thermodynamic stability of various manganese oxide phases as water causes variations in crystal lattice and consequently in electrical conductivity and electrode potential [28,32]. Moreover, water creates challenges in lithium battery application because of lithium instability with water and decomposition of water during charge/discharge processes.

Specific capacity of the electrode after annealing at  $100^\circ\text{C}$  is lower than that annealed at  $300^\circ\text{C}$ . Hydrated water content affects lithium-ion storage in the manganese oxide electrode; therefore an electrode after  $300^\circ\text{C}$  annealing has the highest specific capacity in all discharging current densities, because water has largely been removed by this high temperature. For temperature higher than  $500^\circ\text{C}$ , the deposited manganese oxide decomposes to  $\text{Mn}_2\text{O}_3$ . A structural change such as this may affect the storage of lithium ions, therefore decreasing its specific capacity.

Fig. 6 shows the effect of electrochemical deposition current density on the specific capacity of manganese oxide. Specific capacity of the manganese oxide electrode deposited at high-current density of  $0.125 \text{ mA cm}^{-2}$  is higher than that of low-current density of  $0.025 \text{ mA cm}^{-2}$ . As mentioned previously, morphology of the manganese oxide changes with the current density during deposition. Electrode deposited at high-current density of  $0.125 \text{ mA cm}^{-2}$  consists of nanowires with diameter  $12\text{--}16 \text{ nm}$  distributed randomly on the SS substrate. These distributed nanowires facilitate the storage of lithium ions, possibly, due to both the increased active surface area and the shortened diffusion path. An electrode deposited at low-current density of  $0.025 \text{ mA cm}^{-2}$  has nanowires in clumps which lead to an increased diffusion resistance. It is generally believed that redox reactions in manganese oxide electrode involve lithium-ion intercalation into the electrode, while lithium ions diffusion (mass-transfer resistance) controls the

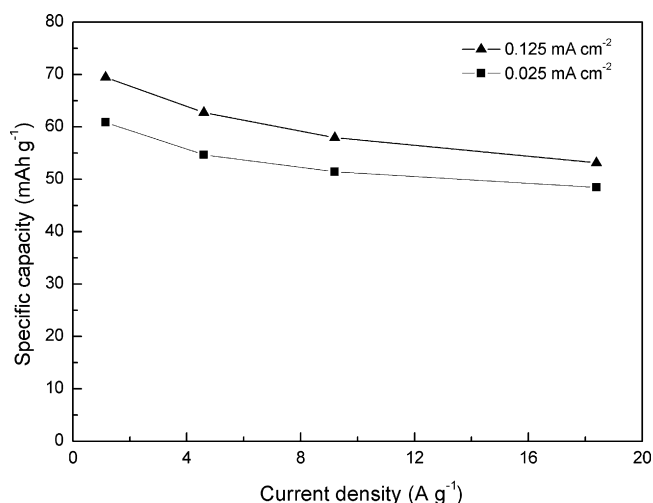


Fig. 6. Effect of current density during electrochemical deposition on the specific capacity of the deposited manganese oxide under different discharging current densities. Electrodes were annealed at 300 °C for 1 h.

specific capacity at high-rate discharging. Therefore, an electrode with distributed nanowires has the highest specific capacity throughout the entire measured current density range because these nanowires shorten the lithium diffusion and in favor of high-rate charging/discharging. In addition, electrode of high-depositing current has a highly porous structure for electrolyte access, which also improves the high-rate discharge capability.

Fig. 7a shows the effect of annealing temperature on cycle-life stability of the deposited manganese oxide electrode. Charging and discharging current density are set at 4.6 A g<sup>-1</sup>. For electrodes annealed below 300 °C, a poor stability is observed: specific capacity decreases from 62 mAh g<sup>-1</sup> to 50 mAh g<sup>-1</sup> after 150 charging/discharging cycles. Hydrated water content is believed to affect cycle-life performance. From Fig. 7a, an enhanced cycle-life performance is seen with increasing the annealing temperature. However, electrodes annealed at temperatures exceeding 300 °C have very little difference in their

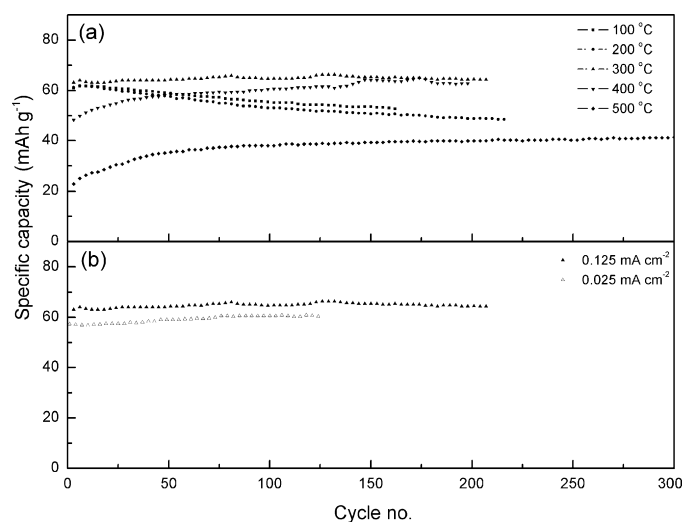


Fig. 7. Effect of annealing temperature (a) and deposition current density (b) on the cycle-life stability of deposited manganese oxide electrode. Charging and discharging current density are set at 4.6 A g<sup>-1</sup>.

cycle-life performance, because most hydrated water content has already been removed. Fig. 7b shows the effect of electrochemical depositing current density on cycle-life stability of the manganese oxides after annealing at 300 °C for 1 h. The effect is small. Therefore, cycle-life stability of a deposited manganese oxide electrode is mainly determined by the annealing temperature rather than the current density during deposition.

#### 4. Conclusion

Electrochemical deposition current density plays an important role in controlling the surface morphology of a deposited manganese oxide film. Morphology of the deposited manganese oxide is changed from random distributed nanowires (about 12–16 nm in diameter) to aggregated nanowire clumps (about 100 nm in diameter) by varying the current density from 0.125 mA cm<sup>-2</sup> to 0.025 mA cm<sup>-2</sup>. Both annealing temperature and current density during deposition affect the electrochemical performance of the deposited manganese oxide materials. Water which impedes lithium intercalation in hydrated manganese oxide is decreased by increasing the annealing temperature. After a threshold temperature of 300 °C, electrodes show a high-specific capacity in all discharging current densities, because most water has already been removed. The structural change caused by decomposition of the deposited manganese oxide at high temperature affects the storage of lithium ions, therefore decreasing its specific capacity. In addition, specific capacity is increased by increasing the deposition current density due to a decreased diameter of the nanowires. Electrode deposited at a high-current density has a high-specific capacity throughout the entire discharge current range because of its small nanowire diameter; narrow nanowires provides more active sites for charge transfer and shortens lithium diffusion, therefore benefits the high-rate charging and discharging.

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